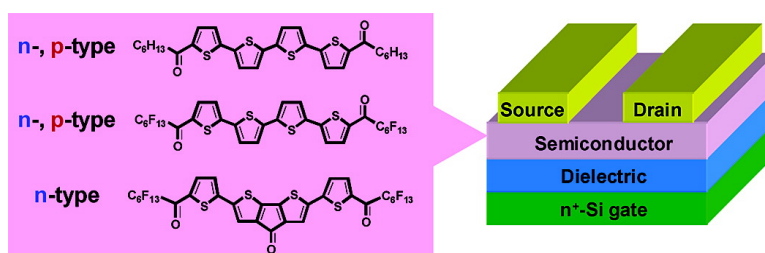


Organic Thin-Film Transistors Based on Carbonyl-Functionalized Quaterthiophenes: High Mobility N-Channel Semiconductors and Ambipolar Transport

Myung-Han Yoon, Sara A. DiBenedetto, Antonio Facchetti, and Tobin J. Marks

J. Am. Chem. Soc., **2005**, 127 (5), 1348-1349 • DOI: 10.1021/ja045124g • Publication Date (Web): 15 January 2005

Downloaded from <http://pubs.acs.org> on March 24, 2009



More About This Article

Additional resources and features associated with this article are available within the HTML version:

- Supporting Information
- Links to the 50 articles that cite this article, as of the time of this article download
- Access to high resolution figures
- Links to articles and content related to this article
- Copyright permission to reproduce figures and/or text from this article

[View the Full Text HTML](#)

Organic Thin-Film Transistors Based on Carbonyl-Functionalized Quaterthiophenes: High Mobility N-Channel Semiconductors and Ambipolar Transport

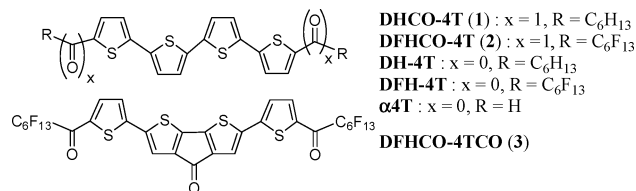
Myung-Han Yoon, Sara A. DiBenedetto, Antonio Facchetti,* and Tobin J. Marks*

Department of Chemistry and the Materials Research Center, Northwestern University, 2145 Sheridan Road, Evanston, Illinois, 60208

Received August 12, 2004; E-mail: a-facchetti@northwestern.edu; t-marks@northwestern.edu

The search for high-performance, environmentally stable electron-transporting (n-type) materials for organic field-effect transistors (OFETs) has just begun.^{1,2} Despite the latest impressive results,² these materials are, compared to hole transporters (p-type),³ far less developed in terms of understanding fundamental structure–charge transport properties relationships and carrier activation–deactivation mechanisms. Furthermore, the availability of n-type (semi)conductors that approach or surpass the carrier mobilities of the most efficient p-type materials³ is important for applications in many organic device technologies⁴ including, but not limited to, OFETs and complementary circuits.^{1–4}

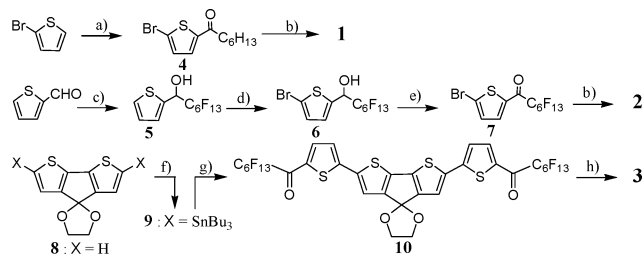
Our group is pursuing this goal using the strategy of selectively functionalizing and/or replacing oligothiophene moieties with perfluorinated substituents.^{2e,5} This approach afforded the first n-type sexithiophene-based OFETs^{5c} and progressive increase of oligothiophene electron mobilities (in $\text{cm}^2 \text{V}^{-1} \text{s}^{-1}$, measured in vacuo) from 0.02^{5c} to 0.08^{2e} and recently to 0.22.^{5a} However, these and many other n-type materials perform nonoptimally in air. In this communication, we report the synthesis and OFET properties of three new carbonyl-functionalized quaterthiophenes exhibiting improved stability and unique charge transport characteristics.



The design rationale for these new structures is the following. The quaterthiophene core was selected since it exhibits one of the greatest and most reproducible OFET mobilities (μ) and current modulation ratios ($I_{\text{on}}/I_{\text{off}}$) in the oligothiophene family, as both p-(α,ω -dihexyl-4T, **DH-4T**) and n-type (α,ω -diperfluorohexyl-4T, **DFH-4T**) semiconductors.^{5a} Carbonyl functionalization was chosen because (i) it introduces one of the stronger electron-withdrawing (EWD) primary functionalities, (ii) in contrast to other powerful EWD groups such as CN and NO₂, it allows additional modifications and functionalization, (iii) it can be readily incorporated into π -conjugated cores, and (iv) it prevents β -fluoride elimination when inserted between a fluoroalkyl chain and carbanion fragment. The new quaterthiophenes **1–3** were synthesized according to Scheme 1 and were characterized by conventional chemical and physical methods.

Cyclic voltammetry⁶ of **1–3** in THF reveals two reversible one-electron reduction processes (E_1/E_2 (V): **DHCO-4T** $-1.06/-1.47$; **DFHCO-4T** $-0.88/-1.01$; **DFHCO-4TCO** $-0.65/-0.78$), considerably less negative than unsubstituted α 4T ($-1.94/-2.07$ V). Oxidation processes are irreversible. UV–vis/PL data (Figure S1) indicate that the C=O groups are effectively conjugated with the

Scheme 1. Synthetic Route to Semiconductors 1–3



(a) $\text{C}_6\text{H}_{13}\text{COCl}$, AlCl_3 , C_6H_6 , reflux.

(b) Bu_3Sn , $\text{C}_6\text{F}_{13}\text{COCl}$, $\text{Pd}[\text{PPh}_3]_4$, DMF, 100°C .

(c) $\text{C}_6\text{F}_{13}\text{I}$, MeLi, Et₂O, -78°C ; HCl. (d) Br₂, CH₂Cl₂.

(e) MnO₂, CH₂Cl₂. (f) *n*-BuLi, Bu₃SnCl, THF.

(g) **7**, Pd[PPh₃]₄, DMF, 90°C . (h) AcOH–HCl, reflux.

4T core and give rise to substantially red-shifted absorption and emission maxima ($\lambda_{\text{abs}}/\lambda_{\text{em}}$ (nm): **DHCO-4T** 430/530; **DFHCO-4T** 465/550; **DFHCO-4TCO** 545/615) and HOMO–LUMO optical gap reductions (E_g (eV): **DHCO-4T** 2.6; **DFHCO-4T** 2.4; **DFHCO-4TCO** 2.2) vs **4T** ($\lambda_{\text{abs}}/\lambda_{\text{em}} = 391/450$ nm, $E_g = 2.8$ eV). HOMO/LUMO energies ($E_{\text{HOMO}}/E_{\text{LUMO}}$ (eV)) are estimated as follows for **DHCO-4T** ($-6.38/-3.78$), **DFHCO-4T** ($-6.36/-3.96$), and **DFHCO-4TCO** ($-6.39/-4.19$) vs α 4T ($E_{\text{HOMO}}/E_{\text{LUMO}} = -5.79/-2.90$ eV), from the relation $E_{\text{HOMO}} = E_{\text{LUMO}} - E_g$, where E_{LUMO} is estimated from E_1 .^{5b} Note that compared to p-type α 4T, the new systems exhibit a considerably larger depression of LUMO (~ 0.9 – 1.3 eV) vs HOMO (~ 0.6 eV) energies, suggesting that carbonyl functionalization of the core should affect electron transport more than hole transport. In marked contrast, on going from α 4T to **DFH-4T** ($E_{\text{HOMO}}/E_{\text{LUMO}} = -6.19/-3.31$ eV) or **DH-4T** ($E_{\text{HOMO}}/E_{\text{LUMO}} = -5.80/-2.89$ eV) a uniform $E_{\text{HOMO}}/E_{\text{LUMO}}$ shift is observed in agreement with σ -EWD substituent effects.^{5b} Theoretical and experimental studies on carbonyl-functionalized (oligo)heteroaromatics indicate that HOMO energies are less affected than LUMOs, since the latter are more localized on the molecular core.⁷ In marked contrast, the LUMO fully extends to the C=O groups via S_{thiophene} → C=O intramolecular charge transfer, enhancing substituent σ/π -EWD effects. Therefore, greater perturbation of the **1–3** LUMOs is expected, in excellent agreement with the present electrochemical and optical data.

The new carbonyl-substituted oligothiophenes are thermally stable and undergo quantitative sublimation. Thin films are readily grown from the vapor phase under vacuum and have been characterized by XRD (revealing molecular edge-on-substrate growth orientation), scanning electron microscopy, and FET I – V measurements. Full details will be reported separately. Top-contact FET devices were fabricated as reported previously.⁵ Briefly, semiconductors **1–3** (~ 50 nm) were vapor-deposited on HMDS-treated p-doped Si/SiO₂ substrates maintained at deposition temperatures

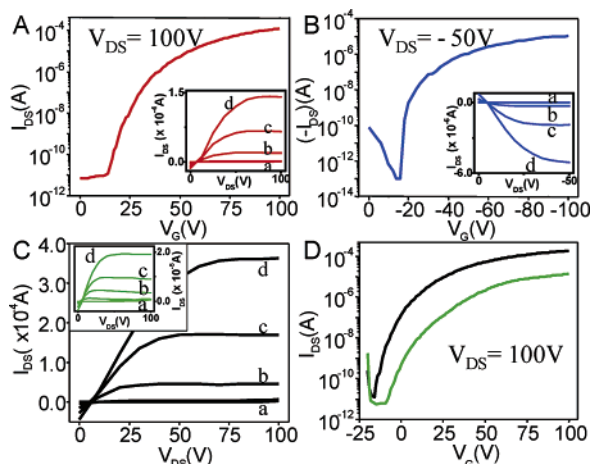


Figure 1. I_{DS} - V plots for: (A) **DHCO-4T** (n-type, vacuum). (B) **DHCO-4T** (p-type, vacuum). (C) **DFHCO-4T** in a vacuum (black) and air (green). (D) **DFHCO-4T** transfer plots in a vacuum (black) and air (green) at different source-drain/gate biases. V_G (V); $a = 0-\pm 40$, $b = \pm 60$, $c = \pm 80$, $d = \pm 100$.

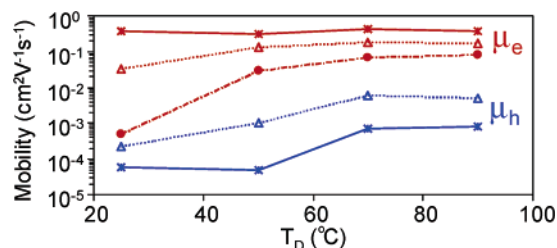


Figure 2. Electron (red) and hole (blue) mobilities vs film deposition temperature (T_D) for **DHCO-4T** (Δ), **DFHCO-4T** (\times), and **DFHCO-4TCO** (\bullet) in a vacuum. μ_e for **DFHCO-4T** is after I_2 vapor treatment.

(T_D) between 25 and 90 °C. OFET fabrication was completed by vapor-depositing source-drain Au contacts (~ 50 nm). Measurements were performed in air and vacuum ($\sim 10^{-5}$ Torr), and the standard saturation FET equation⁸ was employed to calculate carrier mobilities.

Figure 1 shows typical source-drain current–voltage plots for 1–3-based OFETs under different conditions. All of the new oligothiophenes exhibit *very high* electron mobilities (μ_e) in vacuum, with average values as a function of T_D shown in Figure 2. Note that for some **DFHCO-4T** devices, μ_e as high as ~ 0.6 $\text{cm}^2 \text{V}^{-1} \text{s}^{-1}$ has been measured. Interestingly, **DHCO-4T** films also exhibit relatively *large hole* mobilities (μ_h up to 0.01 $\text{cm}^2 \text{V}^{-1} \text{s}^{-1}$ at $T_D = 70$ °C) at all deposition temperatures. Ambipolar transport has been observed previously in blend and bilayers and single component OFETs, but with modest figures of merit, large imbalances between μ_e and μ_h , and only for narrow T_D ranges.^{2f,9} To our knowledge, **DHCO-4T** is the first organic conductor exhibiting unoptimized μ_e/μ_h values as high as $\sim 0.1/0.01$ $\text{cm}^2 \text{V}^{-1} \text{s}^{-1}$ (vacuum, $T_D = 70$ – 90 °C). **DFHCO-4T** also exhibits ambipolar behavior but only after I_2 vapor treatment. From the transfer plots, very high $I_{on}:I_{off}$ ratios are observed for electrons: $> 10^7$. Maximum current gains for holes are $> 10^8$ for **DHCO-4T**. The subthreshold swings (S), indicating how sharply the devices turn on, are in the 1.3–4.9 V/decade range, and turn-on voltages $|V_0|$ are ~ 5 – 30 V.

More electron-deficient **DFHCO-4T** (Figure 1C) and **DFHCO-4TCO** (Figure 1D) also operate well in air, with μ_e 's exhibiting the same T_D dependence as in a vacuum (Figure 2) but with $\sim 5/10\times$ lower magnitudes. In particular, **DFHCO-4TCO**-based devices can be cycled more than 20 times in air without obvious degradation. Finally, since dioxolane-protected 4T **10** is solution-processable

and its films are readily converted to **DFHCO-4TCO** films by deprotection via H_2O – HCl vapor treatment and annealing (Figure S2), we are exploring this approach for depositing n-type materials from soluble precursors.

In conclusion, new carbonyl-functionalized quaterthiophenes have been prepared and characterized. These materials possess low-lying LUMOs that allow and facilitate electron injection and transport as well as HOMO energies compatible with respectable hole transport. OFETs fabricated by conventional methods exhibit the largest oligothiophene thin-film μ_e 's found to date and approach the performance of pentacene/ SiO_2 OFETs.¹

Acknowledgment. We thank ONR (N00014-02-1-0909), the NSF-MRSEC program through the Northwestern Materials Research Center (DMR-0076097), and the NASA Institute for Nanoelectronics and Computing (NCC 2-3163) for support.

Supporting Information Available: 1–3 syntheses and spectroscopic data and FET device fabrication. Figures S1 and S2. This material is available free of charge via the Internet at <http://pubs.acs.org>.

References

- (1) (a) Horowitz, G. *J. Mater. Res.* **2004**, *19*, 1946. (b) Sirringhaus, H. *Nat. Mater.* **2003**, *2*, 641. (c) Rogers, J. A.; Bao, Z.; Katz, H. E.; Dodabalapur A. In *Thin-Film Transistors*; Kagan, C. R., Andry, P., Eds.; Marcel Dekker: New York, 2003; pp 377–425. (d) Dimitrakopoulos, C. D.; Malenfant, P. R. L. *Adv. Mater.* **2002**, *14*, 99.
- (2) (a) Sakamoto, Y.; Suzuki, T.; Kobayashi, M.; Gao, Y.; Fukai, Y.; Inoue, Y.; Sato, F.; Tokito, S. *J. Am. Chem. Soc.* **2004**, *126*, 8138. (b) Kunugi, Y.; Takimiya, K.; Toyoshima, Y.; Yamashita, K.; Aso, Y.; Otsubo, T. *J. Mater. Chem.* **2004**, *14*, 1367. (c) Chesterfield, R. J.; McKeen, J. C.; Newman, C. R.; Frisbie, C. D.; Ewbank, P. C.; Mann, K. R.; Miller, L. L. *J. Appl. Phys.* **2004**, *95*, 6396. (d) Waldauf, C.; Schilinsky, P.; Perisutti, M.; Hauch, J.; Brabec, C. J. *Adv. Mater.* **2003**, *15*, 2084. (e) Facchetti, A.; Yoon, M.-H.; Stern, C. L.; Katz, H. E.; Marks, T. J. *Angew. Chem., Int. Ed.* **2003**, *42*, 3900. (f) Chesterfield, R. J.; Newman, C. R.; Pappenfus, T. M.; Ewbank, P. C.; Haukaas, M. H.; Mann, K. R.; Miller, L. L.; Frisbie, C. D. *Adv. Mater.* **2003**, *15*, 1278. (g) Babel, A.; Janekhe, S. A. *J. Am. Chem. Soc.* **2003**, *125*, 13656. (h) Shim, M.; Javey, A.; Kam, N. W. S.; Dai, H. *J. Am. Chem. Soc.* **2001**, *123*, 11512. (i) Dodabalapur, A.; Lin, Y. Y.; Filas, R. W.; Bao, Z.; LaDuca, A.; Sarpeshkar, R.; Katz, H. E.; Li, W. *Nature* **2000**, *403*, 521.
- (3) (a) Mas-Torrent, M.; Hadley, P.; Bromley, S. T.; Ribas, X.; Tarres, J.; Mas, M.; Molins, E.; Veciana, J.; Rovira, C. *J. Am. Chem. Soc.* **2004**, *126*, 8546. (b) Aramaki, S.; Sakai, Y.; Ono, N. *Appl. Phys. Lett.* **2004**, *84*, 2085. (c) Sundar, V. C.; Zaumseil, J.; Podzorov, V.; Menard, E.; Willett, R. L.; Someya, T.; Gershenson, M. E.; Rogers, J. A. *Science* **2004**, *303*, 1644. (d) Gorjanc, T. C.; Levesque, I.; D'Iorio, M. *Appl. Phys. Lett.* **2004**, *84*, 930. (e) Katz, H. E.; Kloc, C.; Sundar, V.; Zaumseil, J.; Briseno, A. L.; Bao, Z. *J. Mater. Res.* **2004**, *19*, 1995. (f) Mohapatra, S.; Holmes, B. T.; Newman, C. R.; Prendergast, C. F.; Frisbie, C. D.; Ward, M. D. *Adv. Funct. Mater.* **2004**, *14*, 605. (g) Morin, J.-F.; Drolet, N.; Tao, Y.; Leclerc, M. *Chem. Mater.* **2004**, *16*, 4619. (h) Murphy, A. R.; Fréchet, J. M. J.; Chang, P.; Lee, J.; Subramanian, V. *J. Am. Chem. Soc.* **2004**, *126*, 1596. (i) Sherar, C. D.; Jackson, T. N.; Eaton, D. L.; Anthony, J. E. *Adv. Mater.* **2003**, *15*, 2009. (j) Meng, H.; Zheng, J.; Lovinger, A. J.; Wang, B.-C.; Van Patten, P. G.; Bao, Z. *Chem. Mater.* **2003**, *15*, 1778. (k) Afzali, A.; Dimitrakopoulos, C. D.; Graham, T. O. *Adv. Mater.* **2003**, *15*, 2066. (l) Meng, H.; Bendikov, M.; Mitchell, G.; Holgeson, R.; Wudl, F.; Bao, Z.; Siegrist, T.; Kloc, C.; Chen, C.-H. *Adv. Mater.* **2003**, *15*, 1090. (m) Halik, M.; Klauk, H.; Zschieschang, U.; Schmid, G.; Ponomarenko, S.; Kiechmayer, S.; Weber, W. *Adv. Mater.* **2003**, *15*, 917.
- (4) Sze, S. M. *Semiconductor Devices: Physics and Technology*; Wiley: New York, 1985.
- (5) (a) Facchetti, A.; Musherush, M.; Yoon, M.-H.; Hutchison, G. R.; Ratner, M. A.; Marks, T. J. *J. Am. Chem. Soc.* **2004**, *126*, 13859. (b) Facchetti, A.; Musherush, M.; Katz, H. E.; Marks, T. J. *Adv. Mater.* **2003**, *15*, 33. (c) Facchetti, A.; Deng, Y.; Wang, A.; Koide, Y.; Sirringhaus, H.; Marks, T. J.; Friend, R. H. *Angew. Chem., Int. Ed.* **2000**, *39*, 4547.
- (6) Versus Fc^+/Fc (0.54 vs SCE/THF) using 0.1 M TBAPF₆ electrolyte.
- (7) (a) Dal Colle, M.; Cova, C.; Distefano, G.; Jones, D.; Modelli, A.; Comisso, N. *J. Phys. Chem. A* **1999**, *103*, 2828. (b) Ferraris, J. P.; Lambert, T. L. *Chem. Commun.* **1991**, 1268.
- (8) Given by $(I_{DS})_{sat} = (WC_p/2L)\mu(V_G - V_0)^2$, where L ($=100/50$ μm) and W ($=5$ mm) are the device channel length and width, respectively, and C_i (~ 10 nF cm^{-2}) is the insulator capacitance and V_0 is the threshold voltage.
- (9) (a) Rost, C.; Karg, S.; Riess, W.; Loi, M. A.; Murgia, M.; Muccini, M. *Appl. Phys. Lett.* **2004**, *85*, 1613. (b) Meijer, E. J.; de Leeuw, D. M.; Setayesh, S.; van Veenendaal, E.; Huisman, B.-H.; Blom, P. W. M.; Hummelen, J. C.; Scherf, U.; Kadam, J.; Klapwijk, T. M. *Nat. Mater.* **2003**, *2*, 678. (c) Locklin, J.; Shinbo, K.; Onishi, K.; Kaneko, F.; Bao, Z.; Advincula, R. C. *Chem. Mater.* **2003**, *15*, 1404. (d) Dodabalapur, A.; Katz, H. E.; Torsi, L.; Haddon, R. C. *Science* **1995**, *296*, 1560.

JA045124G